



Electron Theory of Ferroelectrics

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Electric Materials



- Dielectrics: insulators
- Piezoelectrics: non-polar/polar w. applied stress/ strain
- Pyroelectrics: polar, spontaneous polarization, unswitchable
- Ferroelectrics: polar, switchable \rightarrow FERAM

Dielectric Constant and Polarization



Electric flux density $D = \varepsilon_0 E$ $|D| = \sigma_f$ Charge $Q = \sigma_f S$ Electric field $|E| = \frac{V}{d} = \frac{\sigma_f}{\varepsilon_0}$

Surface charge density (C/m^2)





 $\begin{array}{c} \textbf{Electric polarization} \\ \textbf{D} = \varepsilon_0 \textbf{E} + \textbf{P} = \varepsilon \varepsilon_0 \textbf{E} & \textbf{P} = \varepsilon_0 \chi \textbf{E} \\ \uparrow & \uparrow & \uparrow & \uparrow \\ \sigma_f & \sigma_p \text{ relative} \\ \textbf{permittivity} & \textbf{Susceptibility} \end{array}$

Dielectric Constant and Polarization



Dielectric constants and electric polarization can be obtained by measuring charge transfer (electric current).

Electric Dipole and Polarization



Localized ion model with point charges $P = \frac{e}{\Omega} \sum_{n} Z_{n}^{*} R_{n}$

inversion operation $r
ightarrow -r \Rightarrow P
ightarrow -P$

no polarization with inversion

effective charge

Paraelectrics and Ferroelectrics



Electron Theory of Ferroelectrics

Ferroelectrics Spontaneous polarization traditional interpretation: polarization of ionic charges



Crucial Factors \rightarrow **Necessity of electron theory**

- Broken inversion symmetry
- Detailed response of electrons

BaTiO₃



Ferroelectric Instability Broken Inversion Why?

displacements $u(Ti) = 0.054 \text{\AA}$ $u(O1) = -0.097 \text{\AA}$ $u(O2) = -0.061 \text{\AA}$

Born-Oppenheimer Potential

- BO Approximation
 - Nucleic system

$$\left[-\sum_{n}\frac{\hbar^2}{2M_n}\nabla_n^2 + \sum_{n>n'}\frac{Z_nZ_{n'}e^2}{|\boldsymbol{R}_n - \boldsymbol{R}_{n'}|} + E_e(\{\boldsymbol{R}_n\})\right]\phi(\{\boldsymbol{R}_n\}) = \mathcal{E}\phi(\{\boldsymbol{R}_n\})$$

BO potential

- Electron system

$$\mathcal{H}_{e}\Psi(\{\boldsymbol{r}_{i}\};\{\boldsymbol{R}_{n}\}) = E_{e}(\{\boldsymbol{R}_{n}\})\Psi(\{\boldsymbol{r}_{i}\};\{\boldsymbol{R}_{n}\})$$

$$\mathcal{H}_e = -\sum_i \frac{\hbar^2}{2m} \nabla_i^2 + \sum_{i>j} \frac{e^2}{|\boldsymbol{r}_i - \boldsymbol{r}_j|} \boxed{-\sum_{i,n} \frac{Z_n e^2}{|\boldsymbol{r}_i - \boldsymbol{R}_n|}}$$
external field by nuclei

- Density Functional Theory
 - One-electron problem in an effective field

Born-Oppenheimer Potential

BO Potential (Total Energy)

$$E(\{\mathbf{R}_n\}) = +\sum_{n>n'} \frac{Z_n Z_{n'} e^2}{|\mathbf{R}_n - \mathbf{R}_{n'}|} + E_e(\{\mathbf{R}_n\})$$

● Atomic Forces → Structure Optimization, MD

$$\boldsymbol{F}_n(\{\boldsymbol{R}_n\}) = -\nabla_n E(\{\boldsymbol{R}_n\})$$

● Force Constants → Phonon, Structure Stability

$$k_{n\alpha,n'\beta}(\{\boldsymbol{R}_n\}) = \frac{\partial^2}{\partial R_{n\alpha}\partial R_{n'\beta}} E(\{\boldsymbol{R}_n\})$$

Phonon Calculation

Atomic Coordinates

$$oldsymbol{R}_n = oldsymbol{R}_{l
u} = oldsymbol{R}_l + oldsymbol{ au}_
u$$

lattice vector coordinates in cell

Atomic Displacements $m{u}_{l
u}=m{R}_{l
u}-m{R}_{l
u}^{(0)}$ equilibrium

positions

Harmonic Approximation with wave vector $oldsymbol{Q}$ and angular frequency ω

$$\boldsymbol{u}_{l\nu} = \boldsymbol{u}_{\nu} e^{i(\boldsymbol{Q} \cdot \boldsymbol{R}_l - \omega t)}$$

Bloch Theorem

$$\boldsymbol{u}_{l'\nu} = \boldsymbol{u}_{l\nu} e^{i\boldsymbol{Q}\cdot(\boldsymbol{R}_{l'}-\boldsymbol{R}_l)}$$

Phonon Calculation

• Atomic Forces

$$F_{l\nu\alpha} = -\sum_{l'\nu'\beta} k_{l\nu\alpha,l'\nu\beta} u_{l'\nu'\beta}$$
$$= -\sum_{\nu'\beta} \bar{k}_{\nu\alpha,\nu'\beta}(\boldsymbol{Q}) u_{l\nu'\beta}$$
$$\bar{k}_{\nu\alpha,\nu'\beta}(\boldsymbol{Q}) = \sum_{l'} k_{l\nu\alpha,l'\nu'\beta} e^{i\boldsymbol{Q}\cdot(\boldsymbol{R}_{l'}-\boldsymbol{R}_{l})}$$

• Equation of Motion

$$-\omega^2 M_{\nu} u_{\nu\alpha} = -\sum_{\nu'\beta} \bar{k}_{\nu\alpha,\nu'\beta}(\boldsymbol{Q}) u_{\nu'\beta}$$
 mass

Phonon Calculation

• Dynamical Matrix

$$D_{\nu\alpha,\nu'\beta}(\boldsymbol{Q}) = (M_{\nu}M_{\nu'})^{-1/2} \bar{k}_{\nu\alpha,\nu'\beta}(\boldsymbol{Q})$$

• Secular Equation

$$\sum_{\nu'\beta} \left[D_{\nu\alpha,\nu'\beta}(\boldsymbol{Q}) - \omega^2 \delta_{\nu\nu'} \delta_{\alpha\beta} \right] u_{\nu'\beta} = 0$$

• Eigenvalue

$$\omega^2(oldsymbol{Q})\geq 0~~$$
 vibration mode

 $\omega^2(\boldsymbol{Q}) < 0$ soft mode (instability)

BaTiO₃

P4mm $\int_{B^{2}} \int_{B^{2}} \int_{B^{2$

ionic model: Ba²⁺ Ti⁴⁺ O²⁻

$$P_S = \frac{e}{\Omega} \sum_n Z_n^* u_n = 0.16 \mathrm{C/m^2}$$

experiment: $P_s = 0.27$ C/m²

displacements $u(Ti) = 0.054 \text{\AA}$ $u(O1) = -0.097 \text{\AA}$ $u(O2) = -0.061 \text{\AA}$

Polarization enhancement Electron response Why?

Macroscopic Polarization R. Martin, Phys Rev B <u>9</u>,1998 (1974)

- Accurate electron density by first-principles calculations
- Sensitive response of electron density spatially distributed due to lattice distortion and ionic displacements
- Electron density: not sufficient to determine polarization

$$oldsymbol{P} = rac{1}{\Omega} \int_{ ext{cell}} oldsymbol{P}(oldsymbol{r}) d^3oldsymbol{r} \qquad
abla \cdot oldsymbol{P}(oldsymbol{r}) = -
ho(oldsymbol{r})$$
 $oldsymbol{P} = rac{1}{\Omega} \int_{ ext{cell}} oldsymbol{r}
ho(oldsymbol{r}) + rac{1}{\Omega} \int_{ ext{surface}} oldsymbol{r} [oldsymbol{P}(oldsymbol{r}) \cdot doldsymbol{S}]$

Theory of Macroscopic Polarization

R. Resta, Ferroelectrics <u>136</u>, 51 (1992).

- Electric dipole is ill defined in a periodic charge distribution unless decomposed into neutral, localized charge segments.
 - The absolute value of polarization *P* is not a bulk property and a variation in *P* is actually observable.

Theory of Macroscopic Polarization

R. Resta, Ferroelectrics 136, 51 (1992).

• Observation of polarization: current by adiabatic variation of state $\lambda \rightarrow \lambda + d\lambda$ $J(\lambda) = \frac{\partial P}{\partial \lambda}$



• First-order polarization variation $\Delta P = \int_{0}^{1} J(\lambda) d\lambda = \int_{0}^{1} \left(\frac{\partial P}{\partial \lambda}\right) d\lambda$

$$\frac{\partial \boldsymbol{P}}{\partial \lambda} = \frac{i\hbar}{m} \frac{e}{\Omega} \sum_{\boldsymbol{k}, i \neq j} \frac{\langle \psi_i^{\boldsymbol{k}}(\lambda) | \boldsymbol{p} | \psi_j^{\boldsymbol{k}}(\lambda) \rangle \langle \psi_i^{\boldsymbol{k}}(\lambda) | \frac{\partial V}{\partial \lambda} | \psi_j^{\boldsymbol{k}}(\lambda)}{\left(E_i^{\boldsymbol{k}}(\lambda) - E_j^{\boldsymbol{k}}(\lambda)\right)^2}$$

$$\mathcal{H}(\lambda)\psi_i^{\boldsymbol{k}}(\lambda) = \left[\frac{\boldsymbol{p}^2}{2m} + V(\lambda)\right]\psi_i^{\boldsymbol{k}}(\lambda) = E_i^{\boldsymbol{k}}(\lambda)\psi_i^{\boldsymbol{k}}(\lambda)$$

Theory of Macroscopic Polarization King-Smith & Vanderbilt, PRB 47, 1651 (1993).

$$\frac{\partial P_{\alpha}}{\partial \lambda} = -\frac{ie}{\Omega} \sum_{\boldsymbol{k},i}^{\text{occ.}} \left[\left\langle \frac{\partial u_{i}^{\boldsymbol{k}}(\lambda)}{\partial k_{\alpha}} \right| \left| \frac{\partial u_{i}^{\boldsymbol{k}}(\lambda)}{\partial \lambda} \right\rangle - \left\langle \frac{\partial u_{i}^{\boldsymbol{k}}(\lambda)}{\partial \lambda} \right| \left| \frac{\partial u_{i}^{\boldsymbol{k}}(\lambda)}{\partial k_{\alpha}} \right\rangle \right]$$

$$\psi_i^{\boldsymbol{k}}(\lambda) = e^{i\boldsymbol{k}\cdot\boldsymbol{r}} u_i^{\boldsymbol{k}}(\lambda)$$

$$\Delta \boldsymbol{P} = \int_0^1 \left(\frac{\partial \boldsymbol{P}}{\partial \lambda}\right) d\lambda = \boldsymbol{P}^{(1)} - \boldsymbol{P}^{(0)}$$

$$\begin{split} P_{\alpha}^{(\lambda)} &= \frac{2ie}{\Omega} \sum_{\mathbf{k},i}^{\text{occ.}} \left\langle u_{i}^{\mathbf{k}}(\lambda) \right| \frac{\partial}{\partial k_{\alpha}} \left| u_{i}^{\mathbf{k}}(\lambda) \right\rangle \\ \\ \mathbf{Berry \ phase} \quad \phi_{\mathbf{k}_{\perp}}^{(\lambda)} &= i \sum_{i}^{\text{occ.}} \int dk_{\alpha} \left\langle u_{i}^{\mathbf{k}(\lambda)} \left| \frac{\partial}{\partial k_{\alpha}} \right| u_{i}^{\mathbf{k}(\lambda)} \right\rangle \end{split}$$

Calculation of Macroscopic Polarization

Variation in polarization

$$\Delta \boldsymbol{P} = \int_0^1 \left(\frac{\partial \boldsymbol{P}}{\partial \lambda}\right) d\lambda = \boldsymbol{P}^{(1)} - \boldsymbol{P}^{(0)}$$

$$\Delta \boldsymbol{P} = \Delta \boldsymbol{P}^{\text{ion}} + \Delta \boldsymbol{P}^{\text{el}}$$

ion=nucleus + core electrons

Localized charges

$$\begin{split} \Delta \boldsymbol{P}^{\rm ion} &= \frac{e}{\Omega} \sum_{\nu} Z_{\nu}^* \boldsymbol{u}_{\nu} \\ & Z_{\nu}^* = Z_{\nu} - N_{\rm core} \quad \text{ionic charge} \end{split}$$

 $oldsymbol{u}_{
u}$ atomic displacement

Calculation of Macroscopic Polarization

Non-overlapping charges

$$\Delta \boldsymbol{P}^{\text{el}} = -\frac{e}{\Omega} \int \boldsymbol{r} \left[n_a^{(1)} - n_a^{(0)} \right] d\boldsymbol{r}$$

With boundary at zero density, integration can be made as a cell integral like localized density.

• Charges with localized orbitals (Wannier ft.)

$$n_{a}^{(\lambda)}(\boldsymbol{r}) = 2 \sum_{n} \left| a_{n}^{(\lambda)}(\boldsymbol{r}) \right|^{2}$$
$$\Delta \boldsymbol{P}^{\text{el}} = -\frac{2e}{\Omega} \sum_{n}^{n} \left(\bar{\boldsymbol{r}}_{n}^{(1)} - \bar{\boldsymbol{r}}_{n}^{(0)} \right) \quad \bar{\boldsymbol{r}}_{n}^{(\lambda)} = \int \boldsymbol{r} \left| a_{n}^{(\lambda)} \right|^{2} d\boldsymbol{r}$$

center of gravity of Wannier ft.

Berry Phase

Berry, PRSL A 392, 45 (1984)

 $|\Psi(t)\rangle = \sum |m(t)\rangle a_m(t)$

Phase associated with adiabatic process originating from geometric property of hamiltonian

Time-dependent Perturbation

$$i\hbar\dot{a}_n(t) = E_n(t)a_n(t) - i\hbar\sum_m \langle n(t)|\dot{m}(t)\rangle a_m(t)$$

Adiabatic Approximation (m=n only)

$$a_n(t) = a_n(0) \exp\left[-\frac{i}{\hbar} \int_0^t E_n(t') dt'\right] \exp\left[i\gamma_n(t)\right]$$
$$\gamma_n(t) = i \int_0^t \langle n(t') | \dot{n}(t') \rangle dt'$$

Berry Phase

- Time dependence via a parameter x(t)
- A one-turn process in the parameter space with time from 0 to T

$$\gamma_n(T) = i \int_0^T \langle n(\boldsymbol{x}(t')) | \dot{n}(\boldsymbol{x}(t')) \rangle dt'$$
$$= i \oint_C \langle n(\boldsymbol{x}) | \nabla_{\boldsymbol{x}} n(\boldsymbol{x}) \rangle \cdot d\boldsymbol{x}$$



Berry Phase of Bloch Function Zak, PRL <u>62</u>, 2747(1989)

1D Bloch function in a vector potential

$$i\hbar\dot{\psi}(x,t) = \left[\frac{1}{2m}\left(p - \frac{e}{c}A(t)\right)^2 + V(x)\right]\psi(x,t) \qquad V(x+a) = V(x)$$
$$\left[\frac{1}{2m}\left(p - \frac{e}{c}A(t)\right)^2 + V(x)\right]\phi_n(x,t) = \varepsilon_n(t)\phi_n(x,t)$$

Berry Phase and Wannier Function

1D Wannier Function

$$a_i(x - X_l) = \frac{1}{\sqrt{N}} \sum_k e^{ik(x - X_l)} u_i(x, k)$$
$$u_i(x, k) = \frac{1}{\sqrt{N}} \sum_l e^{-ik(x - X_l)} a_i(x - X_l)$$
$$\frac{\partial}{\partial k} u_i(x, k) = \frac{1}{\sqrt{N}} \sum_l -i(x - X_l) e^{-ik(x - X_l)} a_i(x - X_l)$$

Berry Phase

$$\begin{split} \gamma_i &= i \int_{-\frac{\pi}{a}}^{\frac{\pi}{a}} \left\langle u_i(k) \Big| \frac{\partial}{\partial k} u_i(k) \right\rangle dk \\ &= \left(\frac{2\pi}{a}\right) \int x |a_i(x)|^2 dx \quad \text{ center of gravity} \end{split}$$

Calculation of Polarization with Berry Phase

Electric Polarization of 3D System

$$\begin{split} \boldsymbol{P}_{el}^{(\lambda)} &= \frac{-2ie}{(2\pi)^3} \sum_n \int d^3 \boldsymbol{k} \langle u_n^{(\lambda)}(\boldsymbol{k}) | \nabla_{\boldsymbol{k}} u_n^{(\lambda)}(\boldsymbol{k}) \\ &= \frac{2e}{(2\pi)^3} \int d^2 \boldsymbol{k}_{\perp} \gamma^{(\lambda)}(\boldsymbol{k}_{\perp}) \end{split}$$



$$\gamma^{(\lambda)}(\boldsymbol{k}_{\perp}) = \operatorname{Im} \left\{ \lim_{J \to \infty} \prod_{s=0}^{J-1} \det S_{ij}^{(\lambda)}(k_s, k_{s+1}) \right\}$$
$$S_{ij}^{(\lambda)}(k_s, k_{s+1}) = \langle u_i^{(\lambda)}(\boldsymbol{k}_{\perp}, k_s) | u_j^{(\lambda)}(\boldsymbol{k}_{\perp}, k_{s+1})$$



Convergency with respect to number of k-points

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Phase Problem of Polarization

$$\frac{eR}{\Omega} \sim \frac{1.6 \times 10^{-19} \text{C} \cdot 4 \times 10^{-8} \text{cm}}{(4 \times 10^{-8}) \text{cm}^3} = 100 \mu \text{C/cm}^2$$



ZnO



Polarization of Ferroelectric BaTiO₃

- Tetragonal (ionic model: 0.16 C/m²)
 - 0.31 C/m² Ishii, TO
 - 0.28 C/m² Zhong, Vanderbilt
 - 0.27 C/m² experiment
- Rhombohedral
 - 0.37 C/m² Ishii, TO
 - 0.43 C/m² Zhong, Vanderbilt
 - 0.33 C/m² experiment

Band Structure of Cubic BaTiO₃



Density of States of Cubic BaTiO₃



Electron Density of Tetragonal BaTiO₃



Phonon Mode of Cubic BaTiO₃



Phonon Band of Perovskite Oxides: ABO₃





Eigenvector: red: A green: B blue: O

Phys. Rev. B 60, 836 (1999)

Energy Variation of BaTiO₃ due to Displacement



Electronic Structure Variation of BaTiO₃ due to displacement





displacement

- → Enhancement in hybridization
- \rightarrow band shift
- \rightarrow energy gain

broken lines: cubic solid lines: tetragonal

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perovskite structure



three-site model

$$\mathcal{H} = \mathcal{H}_0 + \mathcal{H}_t + \mathcal{H}_{ela}$$

$$\mathcal{H}_{0} = \sum_{\sigma} \varepsilon_{d} (d_{1\sigma}^{\dagger} d_{1\sigma} + d_{2\sigma}^{\dagger} d_{2\sigma})$$
$$\mathcal{H}_{t} = t_{1} \sum_{\sigma} \left(d_{1\sigma}^{\dagger} p_{\sigma} + h.c. \right)$$
$$+ t_{2} \sum_{\sigma} \left(d_{2\sigma}^{\dagger} p_{\sigma} + h.c. \right)$$
$$\mathcal{H}_{ela} = \frac{\lambda}{2} x^{2}$$

Variation of hopping integral by O-ion displacement

$$t_1 = -t(1 + \delta \cdot x) \qquad t_2 = +t(1 - \delta \cdot x)$$

electron-lattice coupling

Energy variation up to 4th order of hopping t

$$\Delta E = \left[-\frac{8t^2\delta^2}{\varepsilon_d} + \frac{32t^4\delta^2}{\varepsilon_d^3} + \lambda \right] \frac{x^2}{2} + \frac{32t^4\delta^4}{\varepsilon_d^3} \frac{x^4}{4}$$

Negative coefficient of 2^{nd} order term \rightarrow instability

$$\left|\frac{t}{\varepsilon_d}\right| \ll 1 \qquad \qquad \frac{8t^2\delta^2}{\varepsilon_d} > \lambda \qquad \qquad \text{olarge } \delta \\ \text{o$$

Equilibrium displacement

$$x^{2} = \frac{\varepsilon_{d}^{3}}{32t^{4}\delta^{4}} \left[\frac{8t^{2}\delta^{2}}{\varepsilon_{d}} - \lambda \right]$$

To satisfy the instability condition, appropriately large size of hopping integral t compare to ϵ_d



- large δ
- small λ
- appropriate t

ionic + covalent bonding

Coupling constant δ $t(R) \propto R^{-(l+l'+1)} \approx t(R_0) \left(1 - \frac{l+l'+1}{R_0}x\right)$

• kind of orbital

Iocal structure

$$\mathcal{H}_t = rac{\lambda}{2} x^2$$
 • local structure: tolerance

Tolerance factor of perovskite structure

Elastic constant λ

 $t = \frac{r_A + r_O}{\sqrt{2}(r_B + r_O)} \begin{array}{l} t > 1 \\ t < 1 \end{array} \begin{array}{l} \mbox{Ferroelectric instability} \\ \mbox{of B ion} \\ \mbox{Antiferroelectric instability} \\ \mbox{_{39}} \mbox{of A ion} \end{array}$

Born Effective Charge

	A	B (01 0	2
BaTiO ₃	2.75	7.16	-5.69	-2.11
SrTiO ₃	2.54	7.12	-5.66	-2.00
CaTiO ₃	2.58	7.08	-5.65	-2.00
KNbO ₃	1.14	9.23	-7.01	-1.68
NaNbO ₃	1.13	9.11	-7.01	-1.61
PbTiO ₃	3.90	7.06	-5.83	-2.56
PbZrO ₃	3.92	5.85	-4.81	-2.48
BaZrO ₃	2.73	6.03	-4.74	-2.01

Zhong, Vanderbilt, PRL <u>72</u>, 3618 (1994).





Mechanism of Effective Charge Enhancement





Movement of bond charge due to variation in hybridization between Ti-3*d* and O-2*p* orbitals

Role of Pb

Lone pair

Repulsion of [6s]² or [6sp]²

widely accepted since 70s especially in Chemistry

Covalent bond with O-2p

- 6s=O-2p Cohen (1992-)
- 6*p*=O-2*p* Miyazawa (2000-)

Role of Pb: PbTiO₃



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fully occupied Pb-6s bonding and anti-bonding

occupied Pb-6*p* bonding states only →covalent

Soft Mode in Cubic PbMO₃ and BiMO₃





Summary

- Electron Theory of Ferroelectrics
 - BO Potential
 - QM Representation of Polarization
- Typical Ferroelectric BaTiO₃
 - Ionic and Covalent Bonding
 - Ferroelectric Instability
 - Born Effective Charge

References

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